

Effect of copper addition on glass transition of silicate–phosphate glasses

Justyna Sułowska · Irena Waclawska ·
Magdalena Szumera

Bretsznajder Special Chapter

© The Author(s) 2012. This article is published with open access at Springerlink.com

Abstract Glass transformation effect of mixed SiO_2 – P_2O_5 – K_2O – MgO – CaO – CuO glasses was studied by DSC, XRD, SEM, and Raman spectroscopy methods. The relationship between the parameters characterizing glass transformation effect and an amount of phosphorous and copper forming the glassy structure was discussed. It was shown that an increasing content of phosphorous increased solubility of copper in the structure of the studied glasses which was the result of P–O–Cu bonds formation. Degree of changes of T_g , Δc_p , and time of relaxation values were higher in glasses with higher content of P_2O_5 and CuO . The observed relations were explained on the basis of the local atomic interactions in the structure of glass.

Keywords Glass transformation · Silicate–phosphate glass · CuO

Introduction

Vitreous substances during heating and cooling exhibit glass transformation effect which is induced by relaxation of strains being the consequence of disordered arrangement of atoms forming glass structure. The relaxation of strains in the glass structure taking place at transformation temperature (T_g) is related to a change of such properties as heat capacity, linear and volume expansion coefficients, and viscosity. Parameters characterizing the glass transformation effect

depend on the nature and the number of components forming the glassy structure.

Copper may exist in glass at different oxidation states namely in the form of Cu^{2+} , Cu^+ , and Cu^0 . This element is one of the glass structure modifiers. However, the introduction of excessive amount of copper into glass structure or wrong proportions of copper in relation to other components may facilitate crystallization of formed melt.

According to the literature glasses from $(50-x/2)\text{Na}_2\text{O}$ – $x\text{CuO}$ – $(50-x/2)\text{P}_2\text{O}_5$ system, where x is less than 34 mol% [1], as well as glasses from $50\text{P}_2\text{O}_5$ – $20\text{Na}_2\text{O}$ – 30CuO system [2] were totally amorphous. For glasses from P_2O_5 – Na_2O – CuO system the glass formation region was determined as $\text{P}_2\text{O}_5 \geq 40$, $\text{CuO} \leq 50$, and $\text{Na}_2\text{O} \leq 60$ mol% [3]. Almost all glasses from $x\text{Cu}_2\text{O}$ – $y\text{P}_2\text{O}_5$ – $z\text{MoO}_3$ system, where x , y , and z changed from 0 to 50 mol% were characterized by high glass forming ability with the exception of glasses with low P_2O_5 content where the ratio of $\text{Cu}_2\text{O}/\text{MoO}_3$ was equal to unity because during cooling of their melt crystallized molybdates such as Cu_2MoO_4 and $\text{Cu}_2\text{Mo}_2\text{O}_7$ [4]. Glasses from $(0.70-x)\text{SiO}_2$ – $0.30\text{Na}_2\text{O}$ – $x\text{CuO}$ system where x changed from 0 to 0.2 mol% were completely amorphous [5]. In the case of glasses from B_2O_3 – SiO_2 – Al_2O_3 – CaO – CuO system boron oxide increased solubility of CuO in their structure. 15 mol% of B_2O_3 and 15–25 mol% of CuO in their structure caused crystallization of fine-needle CuO crystals during their synthesis. However, if B_2O_3 content was greater than 20 mol% these glasses did not show any tendency to crystallize during the synthesis [6].

In the literature, we can find information on the effect of copper on the glass transition effect of phosphate and silicate glasses. It was found that addition of increasing amount of CuO at the cost of NaO and P_2O_5 in structure of glasses from $(50-x/2)\text{Na}_2\text{O}$ – $x\text{CuO}$ – $(50-x/2)\text{P}_2\text{O}_5$ system, where x is less than 34 mol% [1], as well as glasses from

J. Sułowska (✉) · I. Waclawska · M. Szumera
Faculty of Materials Science and Ceramics, AGH University
of Science and Technology, Mickiewicza 30,
30-059 Cracow, Poland
e-mail: sulowska@agh.edu.pl

P_2O_5 – Na_2O – CuO system [3] caused increase of the glass transition temperature T_g due to depolymerization of the glass network on smaller units which in turn combined by P–O–Cu bonds and resulted in an increase of crosslinking density of the glasses. According to the literature the ratio of Cu^{2+} ions to total amount of copper ions in the structure of glasses from $50P_2O_5$ – $20Na_2O$ – $30CuO$ [2] and $xCuO$ – $(1-x)P_2O_5$ ($0 \leq x \leq 0.50$) [7] systems had impact on their T_g . The higher the ratio, the higher value of T_g of these glasses.

The introduction of increasing amount of Cu_2O to the structure of glasses from xCu_2O – yP_2O_5 – $zMoO_3$ system where x , y , and z were from 0 to 50 mol% range, resulted in a decrease of the glass transition temperature (T_g) because of formation of weaker ionic bonds Cu^+ –O at the expense of strong covalent P–O and Mo–O bonds [4].

Increase of copper content in the structure of glasses from $(0.70-x)SiO_2$ – $0.30Na_2O$ – $xCuO$ system where x was from 0 to 0.2 mol% resulted in decrease of their T_g [5].

The method of synthesis of glasses from Cu_2O – Na_2O – Al_2O_3 – SiO_2 system also had impact on the value of the glass transition temperature (T_g) [8, 9]. Synthesis of these glasses using Cu^+/Na^+ ion-exchange treatment enabled to regulate the valence of copper ions in the structure of such glasses. The more Cu^+ ions were exchanged in the course of their synthesis by ion exchange, the lower was the value of T_g of these glasses.

However, there was no information on the effect of copper on glass transition of mixed silicate–phosphate glasses.

The object of these studies was silicate–phosphate glasses from SiO_2 – P_2O_5 – K_2O – MgO – CaO – CuO system which can act as slow-dissolving fertilizers providing plant macroelements (P, K, Mg, and Ca) [10], as well as copper acting as a microelement. Copper plays an active role in many life processes of plants, such as photosynthesis,

respiration, metabolism of nitrogen compounds, the formation of proteins, carbohydrate transport, and formation of RNA and DNA [11].

The purpose of this study was to investigate the influence of CuO content on the glass forming ability and glass transition behavior of glasses from SiO_2 – P_2O_5 – K_2O – MgO – CaO system.

Experimental

Two groups of silicate–phosphate glasses from SiO_2 – P_2O_5 – K_2O – MgO – CaO – CuO differing in content of P_2O_5 and CuO were prepared. In both groups constant quantities of K_2O and SiO_2 were kept, and the increasing amount of CuO was introduced at the cost of decreasing amount of MgO and CaO , with the constant MgO/CaO ratio.

The silicate–phosphate glasses were produced by melting raw materials mixture, i.e., SiO_2 , H_3PO_4 , MgO , K_2CO_3 , $CaCO_3$, and CuO at 1450 °C. Then the batch-free glasses were fritted in water. The obtained glasses were grinded to the grain size of 0.1–0.3 mm.

Chemical composition of the synthesized glasses was determined by the X-ray fluorescence spectrometry (XRF) method, using ARL ADVANT'XP spectrometer and was presented in Table 1.

All glasses were subjected to differential scanning calorimetry (DSC) measurements conducted on Netzch STA 449 F1 Jupiter, operating in the heat flux DSC mode. Glass samples weighing 60 mg were heated in platinum crucibles at a rate of 10 °C min^{-1} in a dry nitrogen atmosphere up to 1,100 °C. Characteristic temperatures of the glass transformation effect such as the onset and the end of the glass transformation region, the glass transformation temperature T_g , determined as the midpoint of the c_p , changes in the glass transformation region and changes of specific heat

Table 1 Chemical composition of the investigated silicate–phosphate glasses from SiO_2 – P_2O_5 – K_2O – MgO – CaO – CuO system

	Name of glasses	Content of components/mol%					
		SiO_2	P_2O_5	K_2O	CaO	MgO	CuO
Glasses with low content of P_2O_5	0Cu2P	40.2	2.1	6.1	21.3	30.3	–
	1.5Cu2P	40.6	2	6	20.5	29.3	1.6
	3Cu2P	39.2	2	6.1	20.3	29.2	3.1
	6Cu2P	40.6	1.5	6.1	18.6	27.1	6.1
	12Cu2P	40.8	2	6.1	15.9	23.2	12.1
Glasses with high content of P_2O_5	0Cu6P	42.6	6.5	6.7	23	21.2	–
	2.5Cu6P	42.6	6.4	6.7	22.8	19	2.5
	5Cu6P	43	6.5	6.9	20.3	18.4	4.9
	6.5Cu6P	40	7	6.3	16.8	23.9	6.4
	18Cu6P	41.7	6.1	6.4	14.6	12.9	18.3
	35Cu6P	37.5	5.3	6	7.7	8.4	35.2

(Δc_p) at T_g points were determined applying the Netzsch Proteus Thermal Analysis Program (version 5.0.0.). Approximated relaxation time for the glass transformation $\tau = (T_{\text{end}} - T_{\text{onset}})\beta^{-1}$ where β is the heating rate [12] was evaluated.

The X-ray diffraction method (X'Pert PRO Diffractometer, Philips) was used to confirm the amorphous state of the samples.

The scanning electron microscopy (SEM) was used to find out the microstructure of the analyzed glasses.

The micro-Raman spectroscopy study of glasses was carried out on Raman Jobin-Yvon T-64000 spectrometer, using 514.5 nm line of Ar laser (30 mW) for excitant. The spectra were recorded in the 1,400–200 cm^{-1} range of Raman shifts.

Results and discussion

Glass forming ability

XRD results showed (Fig. 1a) that glasses with higher P_2O_5 content containing less than 18 mol% of CuO were completely amorphous. In the case of higher content of CuO in the structure of these glasses, XRD diffractograms revealed the presence of Cu_2O in their structure.

Results of microstructure studies of the glasses with higher content of P_2O_5 with 6.5 and 18 mol% of CuO showed the presence of spherical inclusions in an amorphous matrix (Fig. 2a, b), which in the presence of higher content of CuO (35Cu6P) take the form of dendritic crystals of Cu_2O (Fig. 2c).

On the other hand, in the case of glasses with a low P_2O_5 content and just 6 mol% of CuO a weak reflex on their diffraction pattern was observed which corresponded to the beginning of copper excess precipitation from the melt (Fig. 1b).

Thus, the higher content of P_2O_5 in the structure of the studied glasses, the higher solubility limit of CuO. This behavior suggested that copper in the silicate-phosphate glass structure formed Cu–O–P bonds.

Effect of copper as the glass modifier on structure of glasses with higher content of P_2O_5 was evidenced by Raman spectra (Fig. 3). They were characterized by bands at 960 and 1,070 cm^{-1} which could be assigned to Si–O[−] stretching vibrations in Q^2 , Q^1 (at 960 cm^{-1}), and Q^3 (at 1,070 cm^{-1}) units in the silicate network. The effect of copper addition on this part of the spectrum was not very noticeable. On the other hand, the very weak band at 454 cm^{-1} for 0Cu6P glass sample could be attributed to symmetric stretching of O–P–O bending modes of the orthophosphate PO_4^{3-} units (Q^0) which shifted towards lower wavenumber with the increasing content of CuO in

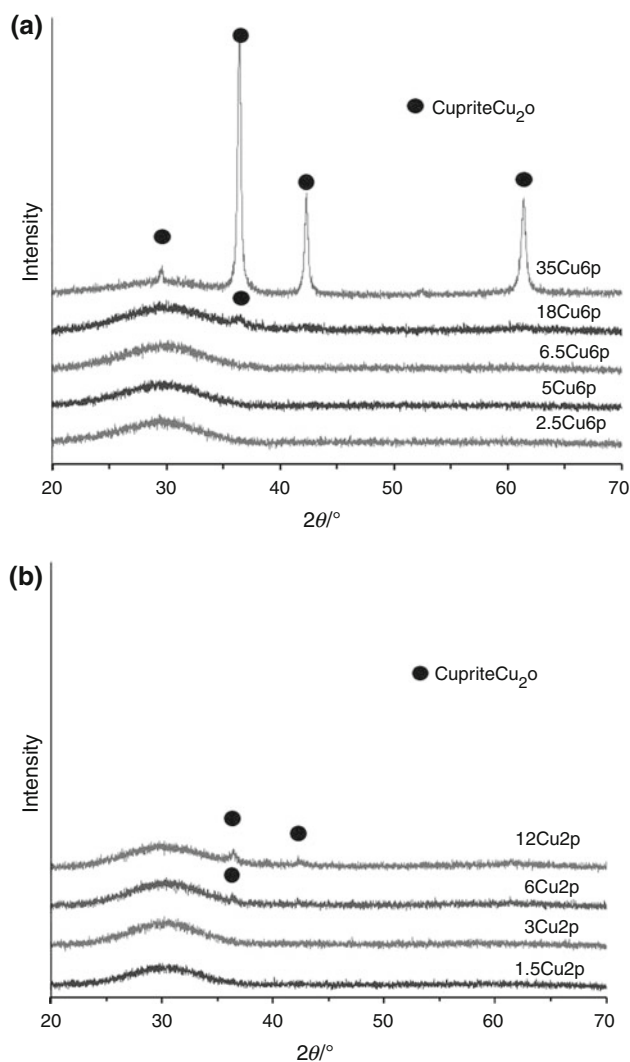


Fig. 1 XRD patterns of the glasses containing; **a** 6 mol% P_2O_5 and **b** 2 mol% P_2O_5

the glass structure, indicating depolymerization of the phosphate network [13].

Sharper than the previous one band at 588 cm^{-1} for 0Cu6P glass sample could be attributed to the symmetric stretching of P–O[−] bonds of the orthophosphate units (Q^0). This band become more distinct with the increasing content of CuO in the structure of analyzed glasses.

The above observations confirmed that some changes around P–O–P bonds in the structure of the considered glasses took place. They might be related to formation of P–O–Cu bonds. Formation of such bonds can be explained based on the difference of the ionicity of bonds in the oxygen bridge (Δi_G), which can be used to measure local chemical affinity of cations to the oxygen in oxide bridge. The higher is its value, the greater is the local chemical affinity of cations to the bridging oxygen [14].

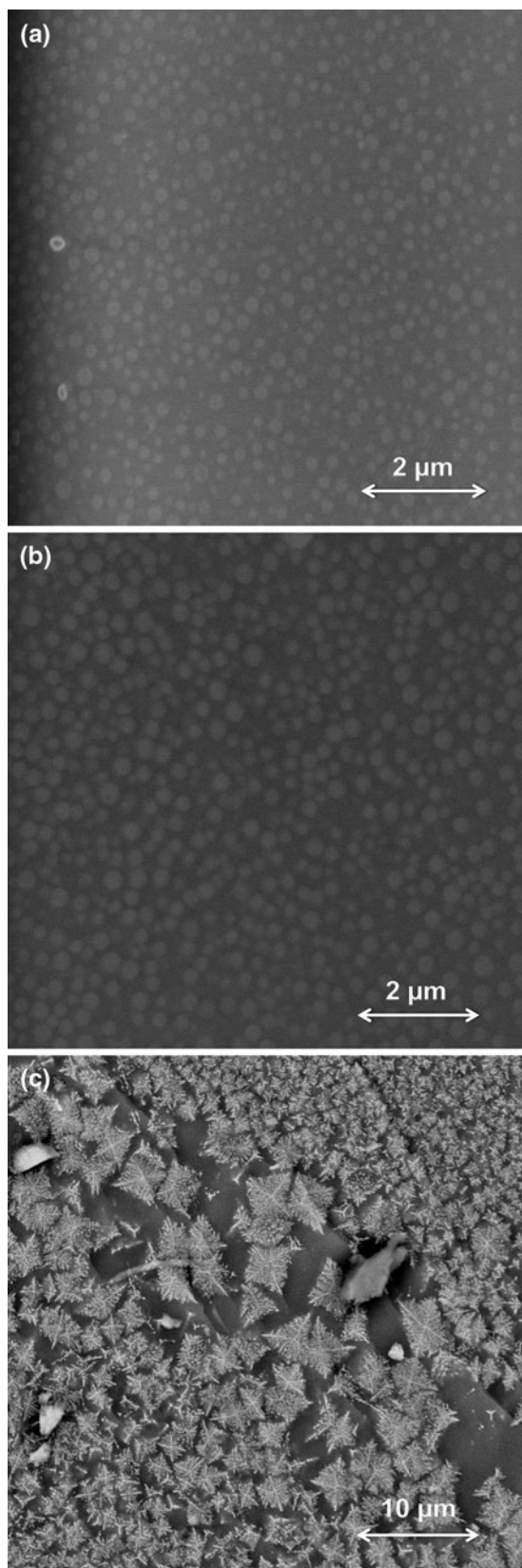


Fig. 2 SEM microphotographs of; **a** 6.5Cu6P, **b** 18Cu6P, and **c** 35Cu6P samples

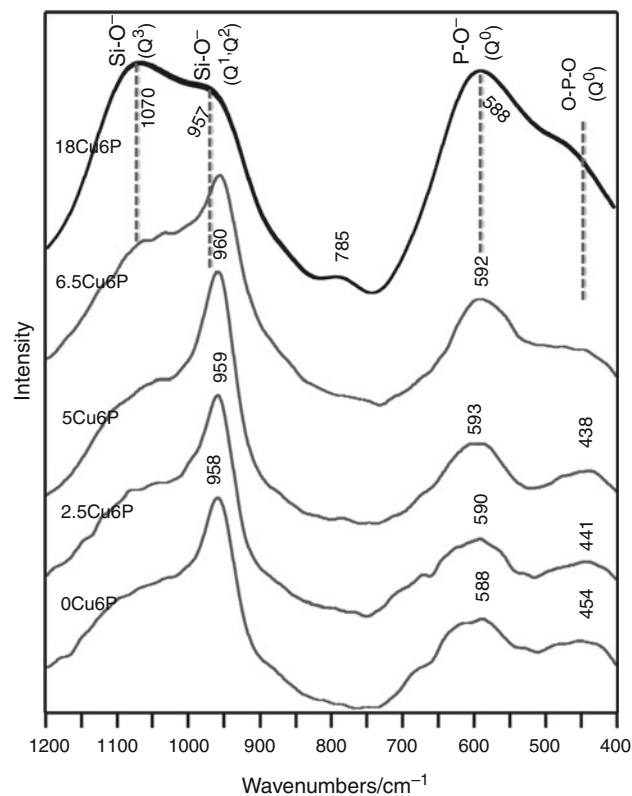


Fig. 3 Raman spectra of the silicate–phosphate glasses with 6 mol% P_2O_5 content

Thus, the presence of copper in the structure of the studied glasses caused a break of P–O–P bonds rather than Si–O–Si bonds due to its greater affinity to phosphorus than to silicon ($Cu^{2+}-O-P \Delta i_G = 0.303 > Cu^{2+}-O-Si \Delta i_G = 0.189$, $Cu^+-O-P \Delta i_G = 0.449 > Cu^+-O-Si \Delta i_G = 0.335$).

Thermal studies

DSC curves of the studied glasses (Fig. 4a, b) showed transformation and crystallization process induced by the thermal treatment. Values of parameters characterizing the glass transformation were presented in Table 2.

It can be noticed that, the introduction of increasing amount of copper into the structure of completely amorphous network (2.5Cu6P, 5Cu6P, 1.5Cu2P, and 3Cu2P) caused broadening of temperature range of the transformation (i.e., increase of the relaxation time), decrease of the characteristic temperature T_g and increase of Δc_p accompanying the glass transformation effect. Such behavior of glasses was observed in the case of silicate–phosphate glasses containing an increasing amount of Fe_2O_3 [15] and B_2O_3 [16], and it was the indication of an increasing degree of their structure relaxation.

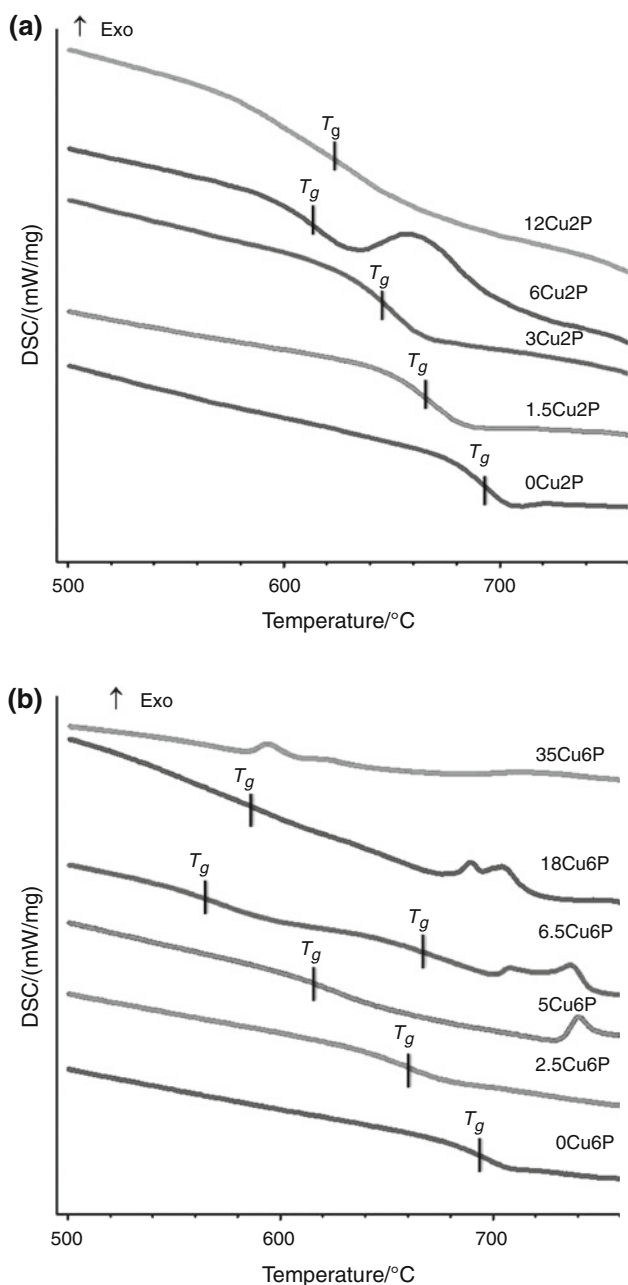


Fig. 4 DSC curves of the analyzed glasses with; **a** 2 mol% P₂O₅ and **b** 6 mol% content

On the DSC curve of glass containing 6.5 mol% of CuO (6.5Cu6P) the double endothermic glass transformation deflection (Fig. 4b) was visible. It was related to phase separation process (liquation) of amorphous matrix of the glass, which was observed in SEM investigations (Fig. 2a, b), and which proceeded the crystallization of Cu₂O (Fig. 2c).

The structural relaxation of the partially recrystallized glasses was accompanied by changes of the specific heat

Table 2 Thermal parameters of the glass transition effect of the analyzed glasses

Name of glasses	T _{onset} /°C	T _g /°C	T _{end} /°C	ΔT _g = T _{end} - T _{onset} /°C	τ = ΔT _g β ⁻¹ /min	ΔC _p /J g ⁻¹ K ⁻¹
Glasses with low content of P ₂ O ₅						
0Cu2P	676	690	701	25	2.5	0.260
1.5Cu2P	651	669	685	34	3.4	0.353
3Cu2P	629	650	663	34	3.4	0.357
6Cu2P	592	609	629	37	3.7	0.303
12Cu2P	577	625	654	77	7.7	0.049
0Cu6P	679	697	705	26	2.6	0.305
2.5Cu6P	639	660	674	35	3.5	0.309
5Cu6P	602	622	650	48	4.8	0.321
6.5Cu6P	547	569	592	45	4.5	0.266
18Cu6P	655	673	696	41	4.1	0.273
35Cu6P	481	592	669	188	18.8	0.060
Glasses with high content of P ₂ O ₅						

ΔC_p which values diminished (18Cu6P) and finally completely disappeared (35Cu6P).

The mentioned above changes of the parameters characterizing glass transformation were more visible in the glasses with higher content of P_2O_5 .

Changes in the glass transformation temperature T_g could be explained based on the nature of chemical bonds in the structure of glasses. The ionicity (i_G) value of bonds of the component atoms with oxygen according to Görlich's scale [17] were applied as a parameter characterizing the strength of bonds. The more covalent character of $Cu^{+2}-O$ bonds ($i_G = 0.617$) replacing the more ionic bonds such as $Ca-O$ bonds ($i_G = 0.707$) and $Mg-O$ bonds ($i_G = 0.670$) caused the glass structure more rigid which consequence was increasing amount of stress ($L_{Cu+2-O} = 1.973$) in the glass, which relaxation required less energy and hence lower values of T_g but higher value of relaxation time.

Complex multistage crystallization processes induced by the thermal treatment in the multicomponent $SiO_2-P_2O_5-K_2O-MgO-CaO-CuO$ glasses with variable copper content were presented in [18].

Conclusions

The appearance of glass transformation in mixed $SiO_2-P_2O_5-K_2O-MgO-CaO-CuO$ glasses and magnitude of parameters characterizing this effect depended on phosphorous and copper content in the glassy structure. The increasing content of phosphorous caused increase of solubility of copper in the structure of the studied glasses which was the result of formation of $P-O-Cu$ instead of $P-O-P$ bonds. The increasing amount of copper introduced to the amorphous matrix of glasses at the cost of decreasing amount of MgO and CaO was the reason of the increasing degree of internal strains relaxation in the glass structure which was accompanied by the decrease of transformation temperature (T_g) value and the increase of glass transformation parameters such as change of molar heat (ΔC_p) and structural relaxation time.

Acknowledgements The study was supported by the Grant No. N R08 0010 06 of the Polish Ministry of Science and Higher Education.

Open Access This article is distributed under the terms of the Creative Commons Attribution License which permits any use, distribution, and reproduction in any medium, provided the original author(s) and the source are credited.

References

- Chahine A, Et-tabirou M, Elbenaissi M, Haddad M, Pascal JL. Effect of CuO on the structure and properties of $(50-x/2)Na_2O-xCuO-(50-x/2)P_2O_5$ glasses. *Mater Chem Phys.* 2004;84:341–7.
- Shih PY, Chin TS. Effect of redox state of copper on the properties of $P_2O_5-Na_2O-CuO$ glasses. *Mater Chem Phys.* 1999;60:50–7.
- Shih PY, Yung SW, Chin TS. Thermal corrosion behavior of $P_2O_5-Na_2O-CuO$ glasses. *J Non Cryst Solids.* 1998;224:143–52.
- Jamnicky M, Znášik P, Tunega D, Ingram MD. Glass formation and structure in the system $Cu_2O-P_2O_5-MoO_3$. *J Non Cryst Solids.* 1995;185:151–8.
- Mekki A, Holland D, McConville CF. X-ray photoelectron spectroscopy study of copper sodium silicate glass surfaces. *J Non Cryst Solids.* 1997;215:271–82.
- Bobkova NM, Artem'eva SA, Trusova EE. Behavior of copper oxide in silicoborate glazed glasses. *Coat Enamles Glass Ceram.* 2007;64(7–8):264–6.
- Metwalli E, Karabulut M, Sidebottom DL, Morsi MM, Brow RK. Properties and structure of copper ultraphosphate glasses. *J Non Cryst Solids.* 2004;344:128–34.
- Lee J, Yano T, Shibata S, Yamane M. Structure and properties of ion-exchanged $Na_2O-Cu_2O-Al_2O_3-SiO_2$ glass system. *J Non Cryst Solids.* 1997;222:120–4.
- Lee J, Yano T, Shibata S, Nukui A, Yamane M. EXAFS study on the local environment of Cu^+ ions in glasses of the $Cu_2O-Na_2O-Al_2O_3-SiO_2$ system prepared by Cu^+/Na^+ ion exchange. *J Non Cryst Solids.* 2000;277:155–61.
- Stoch L, Stoch Z, Waclawska I. Silicate glass fertilizer. Patent PL 185 229 B1. 2003 [Polish].
- Krzywy E. Fertilization of soils and plants. Szczecin: University of Agriculture; 2000 [Polish].
- Soliman AA, Kashif I. Copper oxide content dependence of crystallization behavior, glass forming ability, glass stability and fragility of lithium borate glasses. *Phys B.* 2010;405:247–53.
- Agathopoulos S, Tulyaganov DU, Ventura JMG, Kannan S, Saranti A, Karakassides MA, Ferreira JMF. Structural analysis and devitrification of glasses based on the $CaO-MgO-SiO_2$ system with B_2O_3 , Na_2O , CaF_2 and P_2O_5 additives. *J Non Cryst Solids.* 2006;352:322–8.
- Stoch L. Crystallochemical aspects of structure controlled processes in oxide glasses. *Optica Applicata.* 2008;XXXVIII(1):225–35.
- Waclawska I, Szumera M. Thermal behaviour of Fe-doped silicate-phosphate glasses. *J Therm Anal Calorim.* 2010;101:423–7.
- Szumera M, Waclawska I, Olejniczak Z. Influence of B_2O_3 on the structure and crystallization of soil active glasses. *J Therm Anal Calorim.* 2010;99:879–86.
- Görlich E. The effective charges and the electronegativity. Kraków: Polish Academy of Art and Science; 1997.
- Sułowska J., Waclawska I., Szumera M., Olejniczak Z. Characterization of thermally induced of crystalline phases in CuO-containing silicate-phosphate glasses. *J Therm Anal Calorim.* 2011. doi 10.1007/s10973-011-1988-5.